

Near-Threshold Sputter Yields of Ruthenium under Argon and Nitrogen Ion Bombardment

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Abstract

Ion surface interactions near sputter-threshold are of interest for various plasma facing materials. We report experimental determination of sputter yields for ruthenium films grown on a quartz crystal microbalance and exposed to Ar^+ and N_2^+ ions in the energy range of 50-300 eV. Comparison to semi-empirical models shows agreement to previously reported yields for argon bombardment. In the case of nitrogen, the Yamamura model was modified to account for molecular effects and the yields are found to be between extremes of rigid and non-rigid molecular approximations proposed by Yao. Ex-situ XPS measurements revealed implantation of nitrogen in the ruthenium film after exposure to nitrogen ions. The discrepancy between the models and experimental results for N_2^+ bombardment is explained by an increase in the surface binding energy of the target leading to a chemically reduced sputter yield.

Keywords:

Sputter threshold, Sputter Yields, Quartz microbalance, Ruthenium, Nitrogen

1. Introduction

1 Terrestrial plasmas occur in a wide range of densities and temperatures.
2 High temperature, dense plasmas in fusion reactors [1], diffuse plasmas in-
3 duced by extreme ultraviolet radiation [2] and dense low pressure plasmas in
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5 laboratory ion sources and for electric propulsion [3]. Plasma facing materials
6 (PFMs), such as divertors in fusion reactors are bombarded by hydrogen and
7 deuterium ion species [4], ion lensing systems for electric propulsion technol-
8 ogy deteriorate under xenon ion impingement and optics in next generation
9 lithography applications face diffuse hydrogen and nitrogen plasmas [5, 6].
10 There is, thus, a need to quantify the damage incurred by various PFMs.
11 Sputter yields provide a reasonable estimate of optic material damage due
12 to mass loss when facing diffuse low energy plasmas. However, for reactive
13 ions, chemical effects become dominant near the sputter threshold and need
14 to be accounted for.

15 While experimental data on sputter yields exist for transition metals under
16 nitrogen ion impingement [7, 8, 9, 10], ruthenium is not well studied.
17 Semi-empirical models account for sputtering by assuming a uniform poten-
18 tial barrier at the surface called the surface binding energy (SBE). This is
19 approximated to the enthalpy of sublimation for most targets. Chemical ef-
20 fects such as compound formation during the sputter process would lead to
21 changes in the SBE. Ruthenium nitride (RuN_x) is predicted to have a posi-
22 tive enthalpy of formation [11]. The formation energy barrier is overcome
23 by the incidence of energetic nitrogen ions or radicals either by magnetron
24 sputtering [12, 13] plasmas or pulsed lasers [14]. The formation of RuN_x
25 during sputter measurements would modify the SBE for such combinations
26 involving reactive species leading to larger deviations in model predictions.

27 This paper serves as beginning to a series of experiments to study the
28 interaction of nitrogen and other reactive ion species to a variety of PFMs.
29 In this report, we consider the sputtering of ruthenium under argon and
30 nitrogen bombardment and ascertain the validity of semi-empirical models
31 and their approximation of the SBE for compound formation.

32 **2. Experimental Details**

33 The setup of the exposure facility has been described in detail in a pre-
34 vious publication [15]. A commercial 15 cm DC Kaufman-type ion source
35 with a molybdenum three-grid lensing system was used for generating mono-
36 energetic ion beams of argon and nitrogen with energies down to 50 eV at
37 current densities of $100 \mu\text{A}/\text{cm}^2$. A residual gas analyser (RGA) was used
38 to measure the background gases in the system, which were dominated by
39 water vapor at a level of 10^{-9} mbar. A faraday cup that also functions as
40 a retarding field energy analyser (FC/RFEA) with an entrance aperture of

41 1.5 mm diameter was retrofitted onto the source. A 300 nm ruthenium film
42 was deposited on an AT-cut quartz crystal microbalance (QCM) by mag-
43 netron sputtering prior to installing it on the source-FC+RFEA assembly.
44 The roughness of the film was not determined. It was shown previously [15]
45 that the sputter yields obtained for a film on a quartz crystal microbalance
46 were similar to the ones determined by exposure of an atomically smooth film
47 on a witness sample. The QCM was placed at the same radial distance from
48 the centre of the source axis as the FC+RFEA. This allowed for the mea-
49 surement of the current variations throughout the exposure of the ruthenium
50 film to the ions.

51 The source was operated at a discharge voltage (V_d) of 50 V and 75 V
52 for argon and nitrogen, respectively. The beam was calibrated using the
53 FC+RFEA and the ion energy distributions after a Gaussian fit were found
54 to be mono-energetic with an energy spread between 5-9 eV over the en-
55 ergy range of interest. Analyses of beam contents from similar Kaufman ion
56 sources provide expectation values for argon and nitrogen: At a V_d of 60 V,
57 the concentration of doubly charged ions in the argon plasma is less than 1%
58 [16]. For nitrogen, experiments have shown [17] that while a wide range of
59 parameters influences the composition of the beam, major fractions in cur-
60 rent result due to N^+ and N_2^+ . Based on their analysis we expect N_2^+ ions
61 making up 80-90% of the beam current for our working conditions with the
62 remainder being N^+ . Performing a weighted average on the yields predicted
63 by the Yamamura model (equation 1), for the various species present in the
64 beam, we estimate the deviation of the yield to an upper limit of 18% for ni-
65 trogen and 9% for argon at 50 eV due to presence of atomic ions and doubly
66 charged ions respectively. We also estimate a reduction of current density
67 measured at the FC due to resonant neutralization processes [18] up to 2 and
68 5 percent for argon and nitrogen respectively at a working distance of 7 cm.

69 Data acquisition of the currents and energy measurements were carried
70 out via a Keithley 2100 micro-ammeter controlled by a MATLAB script
71 running on a personal computer. The QCM frequency response and the ion
72 source were controlled through a LABVIEW program.

73 The frequency response from the QCM was converted to a thickness value
74 using the Z-match method [15, 19]. Data between the distortions induced by
75 thermal shock at the start and end of the ion irradiation was used to calculate
76 rate of thickness loss (Figure 1) and subsequently, the sputter yields.

77 The current fluctuations over the dose were estimated to be under 5% for
78 the energies of interest. The exposure energies were selected at random to

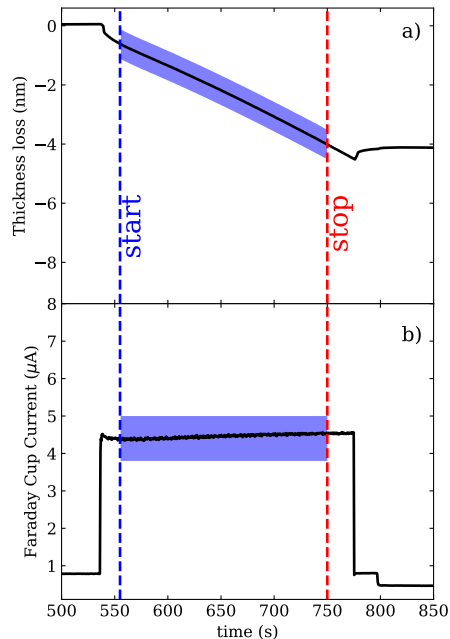


Figure 1: Measurement of thickness loss (a) and ion current (b) made simultaneously with a quartz crystal microbalance and a Faraday cup. The start and stop lines (representative purposes) denote the data used for determining the sputter yield.

79 avoid systematic errors. In both experiments, the surface oxide was sputter
 80 cleaned with 300 eV Ar^+ . Additionally, for nitrogen exposures, the surface
 81 was treated with 300 eV Ar^+ after each exposure to obtain a clean ruthenium
 82 surface prior to each nitrogen ion exposure. Each experiment was carried
 83 out until a dose of 1×10^{18} ions/ cm^2 was irradiated on the surface to ensure
 84 sputter yield is measured at steady state and enough statistics are available
 85 for the low energy exposures.

86 At the end of the experiment, the ruthenium coated QCM sample was
 87 transferred to an X-Ray Photoelectron Spectroscopy (XPS) setup for charac-
 88 terization. The sample remained in atmosphere for two days before it could
 89 be characterized. XPS measurements were done for the ruthenium film at
 90 the last energy of irradiation: 60 eV nitrogen ions, using a Thermo-Fisher
 91 Theta probe instrument with a beam spot size of 400 μm diameter at an
 92 angular range of $\pm 30^\circ$ around an average take-off angle of 53° with respect
 93 to the surface normal.

94 **3. Theoretical Models**

95 We choose the Yamamura [20] model for comparison with experimental
 96 data because of its extensive usage, and Ecksteins [21] formula for its imple-
 97 mentation simplicity. Both build upon Sigmunds original sputter formula to
 98 take into account the presence of a sputter threshold. Ecksteins model pro-
 99 vided modifications over the Yamamura model by correcting the steep rise
 100 of yields above the threshold energy otherwise predicted by the Yamamura
 101 model. This was shown to agree well for inert gas ions and self sputtering
 102 [22]. It is applied for a wide range of ion-target combinations [23, 24]. The
 103 Yamamura model is described as:

$$Y(E) = 0.042 \frac{Q(Z_2)\alpha^*\left(\frac{M_1}{M_2}\right)}{U_s} \frac{S_n(E)}{1 + \Gamma k_e \epsilon^{0.3}} \left(1 - \sqrt{\frac{E_{th}}{E}}\right)^s \quad (1)$$

104 where M_1 is the mass of the ion, M_2 is the mass of the target atom; E_{th} ,
 105 s and Q are fitting parameters. E_{th} is the threshold energy for sputtering.
 106 S_n is the nuclear stopping power of the target element and the parameter Γ
 107 factors in the contribution of reflected ions to the recoil cascade and U_s is the
 108 SBE which is usually approximated to the sublimation energy of the target
 109 material. And Ecksteins formula takes the form:

$$Y(E) = q s_n^{KrC}(\epsilon_L) \frac{\left(\frac{E}{E_{th}} - 1\right)^\mu}{\frac{\lambda}{w(\epsilon_L)} + \left(\frac{E}{E_{th}} - 1\right)^\mu} \quad (2)$$

110 where q , E_{th} , μ and λ are used as fitting parameters. s_n^{KrC} is the reduced
 111 nuclear stopping power based on the Kr-C interaction potential which is a
 112 function of the reduced ion energy, ϵ_L . ϵ_L scales the ion energy with the
 113 Lindhard screening length and the masses of the interacting species. Refer
 114 to [22] for details. In both cases, Y is the theoretical sputter yield and E is
 115 the energy of the incident ion.

116 While these formulae give a reasonable estimate to the sputter yields for
 117 noble gas ions, their accuracy for nitrogen is unreliable due to poor compar-
 118 ison to experimental data. The models have, thus far not been compared
 119 with the present ion-target combination. For reactive species, their predic-
 120 tions can be considered a static case, i.e., the yield observed when the first
 121 ion hits the target. This would lead to an over-estimation when compound
 122 formation occurs, and an under-estimation for chemical sputtering processes.

123 The SBE would deviate dynamically as the surface composition and ion flu-
124 ence changes. Compound formation or accumulation of ions would affect the
125 sputter yield [9, 10]. The models are limited in their ability to account for
126 changes in composition of target materials due to implantation or compound
127 formation. The Yamamura model provides a surface binding energy term to
128 include a surface potential for sputtering. In order to assess chemical changes
129 by the ion species, a fit to the Yamamura model was performed where the
130 SBE (U_s) was added as a fit parameter.

131 4. Results and Discussion

132 Figure 2a shows the sputter yields obtained for ruthenium under argon
133 bombardment. The yields obtained for argon are consistent with data re-
134 ported by Laegreid [25] for polycrystalline rods and by Wu [15] for e-beam
135 deposited ruthenium. The deviations in the data can be accounted by the
136 changes in film density and roughness in the targets studied [26, 27]. A
137 Markov Chain Monte Carlo (MCMC) code [28] was used to calculate the
138 posterior probabilities of the fitting parameters and only the present dataset
139 was used for ease of comparison with the nitrogen data and prevent any bias
140 by the additional datasets. The fit result is plotted in Figure 2a. The ob-
141 tained best-fit value for U_s is 7.5 ± 2.8 eV which is similar to the sublimation
142 energy of 6.7 eV used for the initial Yamamura prediction. A sputter thresh-
143 old of 38.4 ± 1.4 eV obtained from the fit is found to be larger than previous
144 work by $\sim 15\%$.

145 Sputter yields from nitrogen bombardment of ruthenium compared to
146 the Yamamura model are shown in Figure 2b. The expected sputtering by
147 molecular species depends on an effective mass, m^* , that lies between two
148 extremes [29] $M_1 \leq m^* \leq 2M_1$ for when the molecule acts as a rigid ensem-
149 ble with twice the mass of a single atom ($2M_1$) or as two individual atoms
150 (M_1) with half of the original energy. The experimental yields in an ideal
151 situation of no chemical interaction, would lie between these two limiting
152 cases. This molecular effect is dominant for when the vibrational frequency
153 of the molecule is comparable to the interaction time of the collision. From
154 Yao's assumptions [29] of ion-surface interaction times, the molecular effect
155 would occur below 250eV for nitrogen ions. For high energies, atomic and
156 molecular species would sputter equally efficiently. For the comparison with
157 experimental data of nitrogen, parameters in equation 1 were modified to
158 account for the two mass limits and depicted in Figure 2b. Using the same

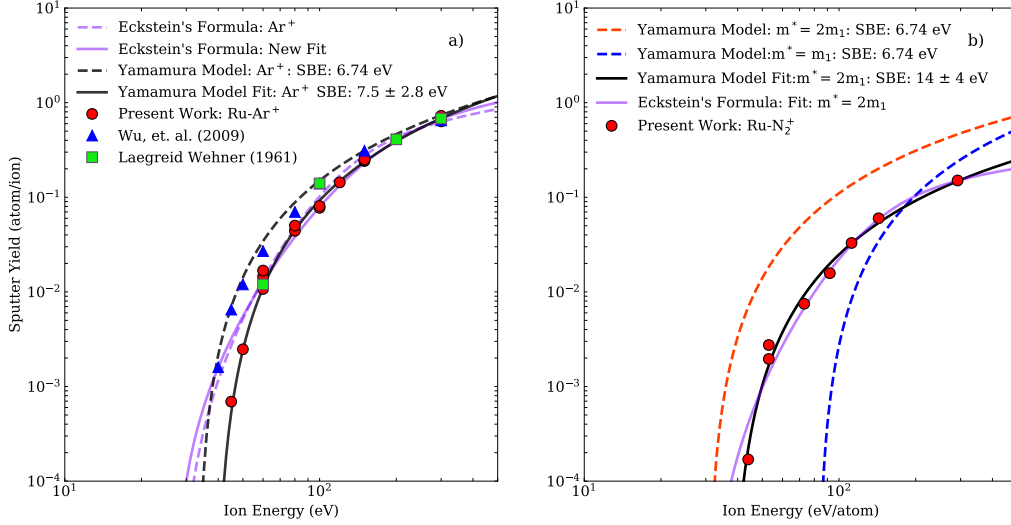


Figure 2: a) Sputter yields obtained from QCM response (circles) compared to reports from Wu (triangles) and Laegreid (squares). The Yamamura (solid black) and Eckstein (solid purple) models with best-fit parameters from Table I are plotted. (b) For N_2^+ yields, the Yamamura model for a rigid ($m^* = 2m_1$) and a non-rigid ($m^* = m_1$) molecule are plotted and compared to a new fit with a modified surface binding energy; New parameters obtained for Eckstein's formula (purple) are also plotted which show marked deviations near the sputter threshold.

159 procedure for fitting the data as for argon, and using the model with m^*
 160 $= 2M_1$, U_s is found to be 14 ± 4 eV. Q and s remain the same within error
 161 for both ion species. A sputter threshold of 37 ± 2 eV was determined from
 162 the fit. Eckstein provides a compendium of tabulated values in [24] for the
 163 fitting parameters for a wide range of ion and target combinations. However,
 164 to our knowledge, no reported best-fit parameter values for a ruthenium and
 165 nitrogen combination are available. We provide an estimate to the model pa-
 166 rameters using the MCMC approach. The results are summarised in Table
 167 I.

168 Chemical interaction between the ruthenium and nitrogen is evident from
 169 the post exposure XPS. The ruthenium 3d spectrum of the QCM exposed to
 170 60 eV nitrogen was compared to a clean ruthenium reference after a Shirley
 171 background subtraction. The contribution to the broadening of the metal
 172 peaks by oxide formation from an exposure to atmosphere and the nitride
 173 formed by the ion exposure experiment are difficult to quantify. The lack of
 174 a significant peak shift of the Ru3d peaks indicates that the sample remains

Ion	Yamamura Model				Eckstein Model				
	Q	s	U_s (eV)	E_{th} (eV)	λ	q	μ	E_{th} (eV)	
Ar^+		1.31 ^a	2.5 ^a	6.74 ^a	33 ^a	0.19 ^b	6.84 ^b	2.20 ^b	27.47 ^b
	MCMC	1.86	2.73	7.5	38.4	0.28	8.4	1.88	26.2
	error	± 0.68	± 0.18	± 2.8	± 1.4	± 0.14	± 1.14	± 0.35	± 5.6
N_2^+		1.31 ^c	2.5 ^c	6.74 ^c	28 ^c	-	-	-	-
	MCMC	1.14	2.75	14	37.2	0.31	1.01	2.13	30.0
	error	± 0.42	± 0.17	± 4	± 2.2	± 0.13	± 0.03	± 0.17	± 4.8

^aReference: [15]

^bReference: [24]

^cReference: [20]

Table 1: Fit parameters with errors obtained from Markov Chain Monte Carlo code after 35,000 steps including a burn-in period of 5000. See [28] for details of the algorithm.

175 mostly metallic, consistent with transition metal nitrides [13]. The O1s peak
176 for air exposed nitrides shares similar difficulties as the contributions from
177 the oxy-nitride and O-H groups from adsorbed water and hydroxides are
178 challenging to deconvolve. The peaks before and after exposure are shown in
179 Figure 3c for qualitative comparison. The oxygen before exposure contained
180 a metallic oxide at around 529eV which is a dominating contribution after
181 exposure to nitrogen and subsequently atmosphere. The quantification of
182 the nitride and oxynitride is therefore limited to the N1s spectra. Two com-
183 ponents of the N1s have been identified, one at 397.3eV corresponding to the
184 nitride and 398.8 eV which is characteristic for an oxynitride for transition
185 metal nitrides exposed to atomsphere [13, 30]. The area under the N1s peak
186 corresponded to 18 atomic percent of nitrogen in the measured sample vol-
187 ume, implying implantation of nitrogen within the target. This is consistent
188 with reports that RuN_x , even though thermodynamically unfavourable, is
189 formed by reactive magnetron sputtering [12].

190 5. Conclusions

191 We report sputter yields for ruthenium under argon and nitrogen ion
192 bombardment. Argon shows agreement to the Yamamura model with dis-

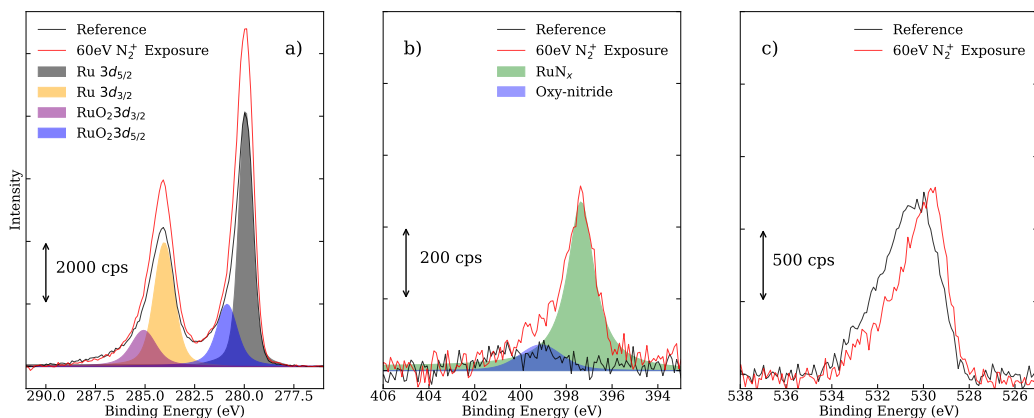


Figure 3: XPS Spectra of Ru coated QCM both measured after exposure to air: with reference (bottom) and 60 eV N_2^+ exposed (top) samples. The Ru3d peaks (a) remain mostly metallic while an N1s (b) peak appears post exposure. See text for details.

193 crepancies compared to literature that possibly originate from density and
 194 roughness variations in the targets analysed. Theoretical yields for nitrogen
 195 ions on Ru provide boundaries for expectation values, but over-estimate ex-
 196 perimental yields for a rigid molecule approximation. SBE is approximated
 197 by sublimation energies for modelling purposes. Transition metals and their
 198 nitrides show variation in the sublimation energy as can be seen in case of Zr
 199 [31] where the experimentally determined sublimation energy of the metal
 200 is 2.7 eV lower than that of the nitride. The approximation of the surface
 201 binding energy to the sublimation energy would then be inappropriate for
 202 a reactive ion bombarding a metal. Using the SBE (U_s) as a fit parameter
 203 showed a change of U_s for ruthenium from argon exposure to nitrogen expo-
 204 sure by 1.8x. We hypothesize that the chemical interaction between nitrogen
 205 and ruthenium, as observed by XPS, leads to formation of ruthenium nitride
 206 and a change in the SBE.

207 Investigation of fluence dependence of yields with an estimation of extent
 208 of nitridation of ruthenium under nitrogen bombardment is planned for an
 209 upcoming series of experiments.

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